

Preparation and deposition of Pr-Fe-B permanent magnet powder using pulsed laser

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We have already prepared a thin permanent magnet with the thickness of sub millimeter by obtaining magnet powders using a PLD (Pulsed Laser Deposition) method. In the study, the PLD followed by a flash annealing enabled us to deposit isotropic Pr-Fe-B magnet powders with coercivity (H_{cj}) > 1000 kA/m on a stainless thin shaft applicable to a miniaturized motor. Observation on the surface of Pr-Fe-B magnets and evaluation on mechanical behavior was carried out. Since the surface of a Pr-Fe-B magnet was coated by a Pr oxide through an annealing process, their magnetic properties didn't degrade after one year. We also confirmed that the Pr-Fe-B magnet has possibility to be applied to a micro magnetization process. It was clarified that the powder technology using the PLD is useful to propose a thin magnet applicable to a next generation small motor.

Index Terms— PLD (Pulsed Laser Deposition), Pr-Fe-B permanent magnet powder, stainless shaft, coating

I. INTRODUCTION

A miniaturized permanent magnet is useful to develop the size reduction of devices in industrial and medical fields [1-3]. Although thinned Nd-Fe-B magnets prepared using a sputtering method have been applied to small electronic devices [4-8], we have difficulty obtaining small Nd-Fe-B magnets with the thickness of sub millimeter due to the low deposition rate. In order to overcome the difficulty, our group has used a PLD method with high deposition rate which is attributed to Nd-Fe-B magnet powders with the size of several microns [9]. Furthermore, small motors comprising sub-millimeter-thick PLD-made Nd-Fe-B magnets such as an axial gap type DC brushless motor and a friction drive motor have been demonstrated.

According to the demagnetization field due to the structure of a small motor aimed at development in future research, increase in coercivity (H_{cj}) up to 1000 kA/m is required because permeance coefficient (P_c) in a demagnetization curve is estimated at 1.0 [10]. Since the magnetic crystalline anisotropy constant of a $\text{Pr}_2\text{Fe}_{14}\text{B}$ phase ($K_u = 6.8 \text{ MJ/m}^3$) is larger by approximately 2.3 MJ/m^3 than that of a $\text{Nd}_2\text{Fe}_{14}\text{B}$ phase ($K_u = 4.5 \text{ MJ/m}^3$), Pr-Fe-B magnets instead of Nd-Fe-B ones were adopted in the study [11].

This contribution reports the deposition of Pr-Fe-B magnet powders with coercivity (H_{cj}) exceeding 1000 kA/m on a thin stainless shaft. Moreover, investigation on various properties was carried out to apply the PLD-made magnet to a miniaturized motor in future.

II. EXPERIMENTAL PROCEDURE

A $\text{Pr}_{2.2}\text{Fe}_{14}\text{B}$ target was ablated with a Nd-YAG pulse laser (wavelength: 355 nm) at the repetition rate of 30 Hz in vacuum atmosphere. The laser power, which was measured with a power meter in front of the entrance lens of a chamber, was approximately 4 W. Before the ablation, the chamber was evacuated down to approximately $2.0\sim 4.0 \times 10^{-5} \text{ Pa}$ by using a rotary pump together with a molecular turbo one. The distance between a target and a stainless shaft (SUS420J2) was fixed at 10 mm. As shown in Fig. 1, A target together with a shaft were rotated during the deposition. In some experiments, Pr-Fe-B powders were deposited on 100 μm -thick stainless (SUS420J2) plates. Average deposition rate exceeded approximately 40

$\mu\text{m/h}$. A pulse annealing (PA) [12] was carried out to crystallize the as-deposited sample.

After the annealing process, samples were magnetized up to 7 T with a pulse magnetizer. Then, J-H loops were measured with a VSM (Vibrating Sample Magnetometer) which could apply a magnetic field up to approximately 2000 kA/m reversibly. The composition of obtained films was evaluated by a SEM (Scanning Electron Microscope)-EDX (Energy Dispersive X-ray spectrometry), and the surface observation was also carried out by using a SEM. An average thickness was measured with a micrometer or estimated by measuring each weight.

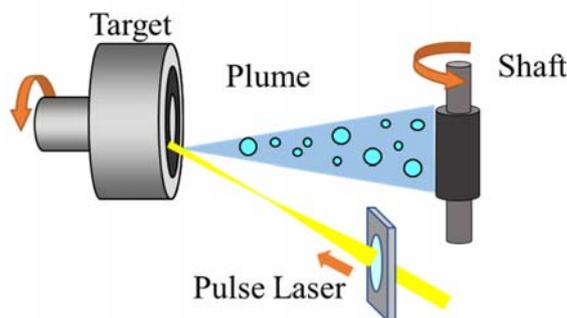


Fig. 1 Schematic diagram of the deposition of Pr-Fe-B powders using a pulsed laser on a thin stainless shaft with the diameter of approximately 0.5 mm.

III. RESULTS AND DISCUSSION

A. Deposition of Pr-Fe-B magnet powders with high coercivity on a thin stainless shaft

Figure 2 shows photos of an isotropic ring-shaped Nd-Fe-B bonded magnet used for a conventional small motor. Overview photo of the 350 μm -thick bonded magnet was displayed as an inset. Surface observation indicated that Nd-Fe-B magnet flakes were bonded using a nonmagnetic phase of thermosetting resin. Comparison of magnetic properties between the conventional bonded magnet and a PLD-made Pr-Fe-B magnet (Pr content= $\text{Pr}/(\text{Pr} + \text{Fe})$: approximately 15 at. %) on a stainless plate was carried out as shown in Fig. 3. The both values of coercivity and residual magnetic polarization of the PLD-made magnet were superior than those of the bonded one. This result

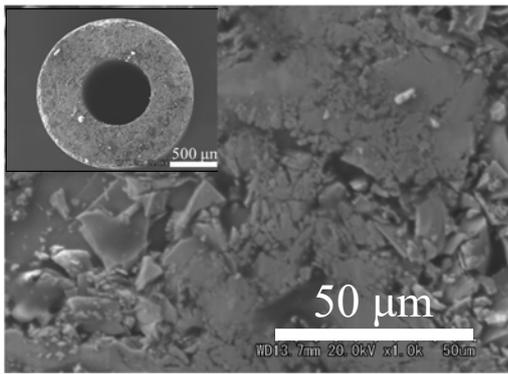


Fig. 2 Photos of a conventional ring-shaped Nd-Fe-B magnet. An overview photo was inserted. As shown in the surface observation, Nd-Fe-B magnet flakes were dispersed in thermosetting resin.

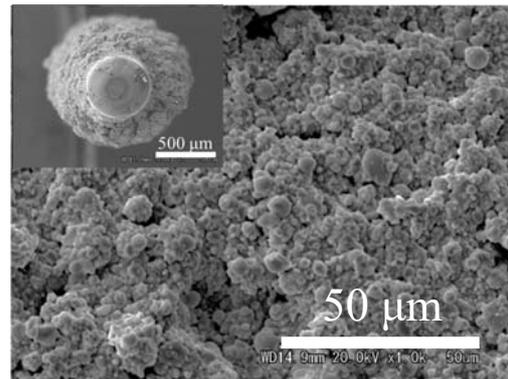


Fig. 4 Schematic diagram of the deposition of Pr-Fe-B powders on a thin stainless shaft. A cross-sectional overview photo was inserted. As shown in the surface observation, Pr-Fe-B magnet powders were consolidated without resin.

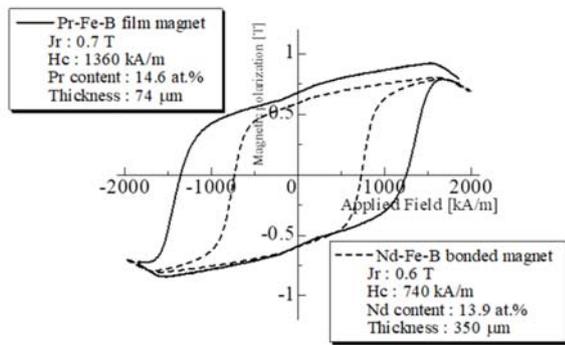
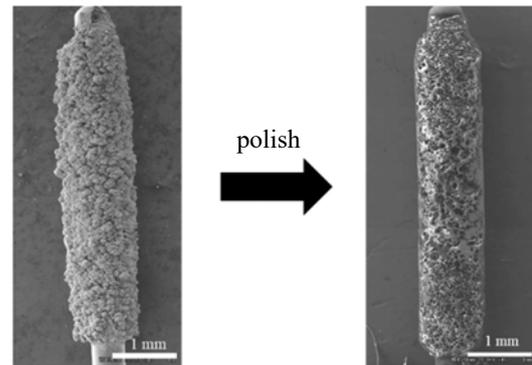


Fig. 3 Two J-H loops of a conventional ring-shaped Nd-Fe-B bonded magnet shown in Fig. 2 together with a PLD-made Pr-Fe-B magnet on a stainless plate. Pr-Fe-B magnet had superior magnetic properties compared with the bonded one.

indicated that the consolidation of the Pr-Fe-B magnet powders using a pulsed laser without nonmagnetic phase was effective to obtain the higher residual magnetic polarization. We, therefore, deposited Pr-Fe-B powders with the same Pr content of approximately 15 at. % on a thin stainless shaft with the diameter of approximately 0.5 mm. (see Figs. 4 and 5 (a)) Mechanical deterioration such as peeling phenomenon didn't occur after a deposition and an annealing process. In addition, the polish of the sample shown in Fig. 5 (b) enabled us to reduce the thickness down to 250 μm and improve the surface roughness as displayed in Fig. 5 (b).

B. Investigation on change with time and thermal exposure of magnetic properties in PLD-fabricated Pr-Fe-B magnet

Evaluation on magnetic properties of PLD-fabricated Pr-Fe-B magnets over one year was carried out to consider a practical application. A Pr-Fe-B magnet with Pr content of approximately 15 at. % on a stainless plate was held in a desiccator with the vacuum of approximately 1 Pa and frequently exposed to the air more than 100 times through one year. Figure 6 shows the change with time of 2nd quadrant of $B-H$ curves in the Pr-Fe-B magnet. A curve of the above-mentioned bonded magnet was also displayed [10]. The $B-H$ curves of each Pr-Fe-B one were completely overlapped. It was clarified that magnetic properties didn't degrade after one year passed. In order to examine the mechanism, the surface of the



(a) after annealing (b) after polishing

Fig. 5 Polish was carried out to reduce the thickness less than 300 μm and improve the surface roughness of an annealed Pr-Fe-B magnet on a stainless shaft.

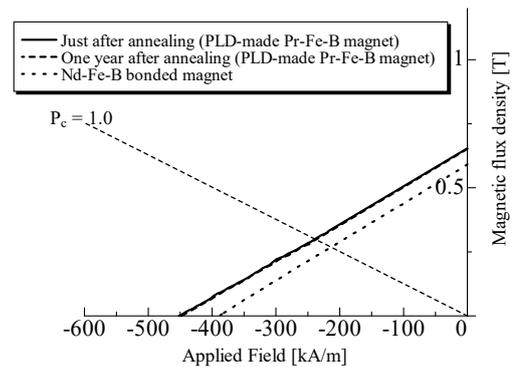
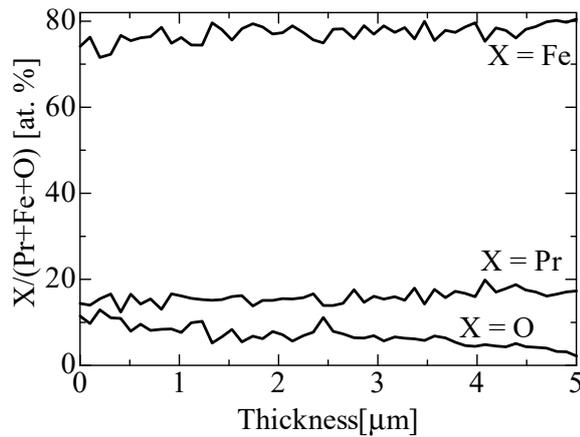
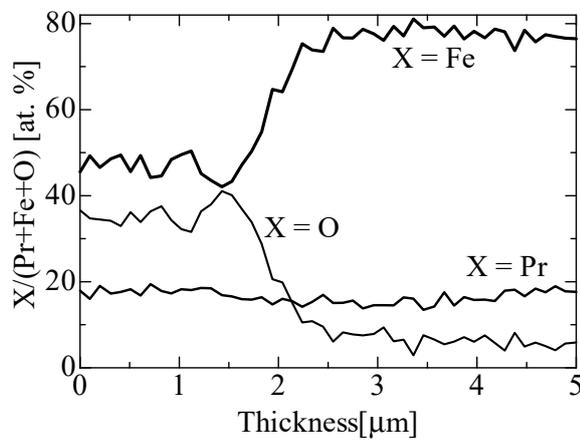


Fig. 6 2nd quadrant of $B-H$ curves of a Pr-Fe-B magnet on a stainless plate and a conventional Nd-Fe-B bonded magnet, respectively. After one year passed, the curve of Pr-Fe-B one overlapped the original one.

sample was observed by using a SEM-EDX as shown in Fig. 7. The horizontal axis means the depth from the surface. The observation revealed that an approximately 2 μm -thick Pr oxide covered the surface of the sample after an annealing process. We suppose that an Pr-Fe-B film was naturally coated by the Pr oxide through an annealing process and the oxidation layer suppressed the deterioration of magnetic properties. In future,



(a) Before annealing (as-deposited)



(b) After annealing

Fig. 7 Composition analysis using SEM-EDX in an as-deposited and an annealed Pr-Fe-B magnet on a stainless plate, respectively. It was confirmed Pr oxide layer was formed in the surface with the thickness of approximately 2 μm through an annealing process.

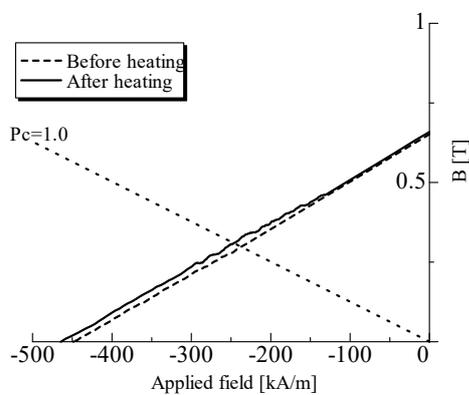


Fig. 8 2nd quadrant of B - H curves of a Pr-Fe-B magnet on a stainless plate before and after a heat treatment at 573 K in the air. The two curves were almost the same.

same observation for a magnet on a thin shaft is required.

Although it has already known that a micro magnetization method is useful to comprise an isotropic (Nd or Pr)-Fe-B magnet to a small motor [13], the magnet is needed to be heated up to approximately 573 K in the air. Figure 8 shows the B - H curves of a PLD-made Pr-Fe-B magnet with Pr content of approximately 15 at. % on a stainless plate before and after the heat treatment. Here, the annealed sample was re-magnetized with a pulse magnetizer. The B - H curves of each Pr-Fe-B one were almost the same. The magnetic properties of the PLD-made magnet didn't degrade after heating at 573 K in the air. It was found that an adoption of micro magnetization for the magnet is hopeful as a future work.

IV. CONCLUSION

Preparation of Pr-Fe-B powders using a PLD enabled us to obtain an isotropic magnet with coercivity (H_c) > 1000 kA/m on a stainless thin shaft without mechanical deterioration. It was also confirmed that each change with time and thermal exposure of magnetic properties is little in the Pr-Fe-B magnets on a stainless plate. These results indicated that the powder technology using the PLD is effective to propose a next generation thin magnet.

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