Bonding of different self-adhesive resins to high-strength composite resin blocks treated with surface conditioning

Type of article: original article

Abbreviated title: bonding to hybrid resin blocks

Keywords: micro-tensile bond strength; contact angle; SEM, EDS, surface roughness

Declarations of interest: none

Hao Yu^{1,2,*}, Keiichi Yoshida³, Hui Cheng¹, Takashi Sawase²

1 Department of Prosthodontics, School and Hospital of Stomatology, Fujian Medical University, China

2 Department of Applied Prosthodontics, Graduate School of Biomedical Sciences, Nagasaki University, Nagasaki, Japan

3 Clinic of Fixed Prosthodontics, Nagasaki University Hospital, Nagasaki, Japan

* Corresponding author at: Department of Prosthodontics, School and Hospital of

Stomatology, Fujian Medical University, China. Fax: +86 591 83720599; E-mail

address: haoyu-cn@hotmail.com (H. Yu)

Number of pages: 24

Number of tables: 6

Number of figures: 2

Quantity of reprints desired: 0

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Abstract

Purpose: This study aimed to evaluate the effects of chemical conditioning and selfadhesive resins (SARs) on the bonding of mechanically conditioned high-strength composite resin block (HSCRB).

Methods: Eighteen sections of HSCRB (KZR-CAD HR 3 Gammatheta, Yamakin) were treated with alumina air abrasion and randomly divided into 3 groups according to the SARs for bonding: RelyX Unicem 2 (RXU), SA Luting Plus (SAL), and G-Cem ONE (GCO). The sections were further divided into 3 subgroups according to the chemical conditioning of the adherend surfaces: no conditioning (C), universal adhesive (UA), and a mixture of γ -MPTS and 10-MDP (MM). After the surface conditioning, the sections were cemented with the SARs. Each cemented section was cut into 40 beams. Half of the beams were thermocycled (4°C/60°C, 10,000 cycles). The micro-tensile bond strength (μ TBS) values were measured using a universal testing machine. Scanning electron microscopy (SEM), energy-dispersive X-ray spectroscopy (EDS), contact angles, and surface roughness measurements were performed on the adherend surfaces of each subgroup.

Results: RXU showed the highest μ TBS values among the 3 SARs tested, while MM application exhibited the highest μ TBS values among the 3 chemical conditioning methods tested. After thermocycling, the samples in the RXU/MM, RXU/UA, and

GCO/MM groups showed no significant changes in the μ TBS values, whereas the others showed a significant reduction.

Conclusions: The bond strength of HSCRB was influenced by the chemical conditioning, SARs, and aging. γ -MPTS and 10-MDP application yielded higher μ TBS values of mechanically treated HSCRB than the UA.

1. Introduction

The global evolution of computer-aided design/computer-aided manufacturing (CAD/CAM) technology led to the development of composite resin blocks (CRBs), which are designed to exhibit the characteristics of conventional ceramics and composites [1, 2]. Currently, two types of CRBs based on microstructure exist: CRBs with dispersed fillers and polymer-infiltrated ceramic network (PICN) materials. With the aid of industrial fabrication with high temperature (>100°C) and/or high pressure (>150 MPa), CRBs were shown to have superior material homogeneity and reliability compared to conventional composites [2]. Contemporary CRBs have a flexural strength ranging from 148.7 to 216.5 MPa and a translucency similar to that of glass ceramics, rendering them an excellent material option for inlays, onlays and veneers [3, 4]. On the other hand, resin-fixed prostheses are covered by Japanese social health insurance and have attracted substantial attention from both patients and clinicians [5, 6].

Adhesive cementation between resin and CRBs is a vital factor in the long-term success of restorations [7]. Schwenter et al. [8] reported that a similar shear bond strength of CRBs was achieved with self-adhesive resins (SARs) and conventional adhesive resins. Because SARs have much lower technical sensitivity, their use is clinically beneficial. Furthermore, since different SARs may exhibit different bonding properties depending on their composition, it is important to carry out a systematic investigation regarding their bonding performances to CRBs. Notably, the high degree of polymerization achieved by the industrial fabrication process may hinder

bonding to CRBs [9, 10]. Therefore, a wide range of surface conditioning methods have been proposed and evaluated by previous studies [4, 6, 11-13], and the choices of conditioning methods are dependent on the compositions of CRBs [14]. The literature seems to reach a consensus that the combination of mechanical conditioning and chemical conditioning prior to bonding yields the highest bond strength of CRBs [4, 11, 13]. More specifically, well-established mechanical conditioning (e.g., alumina air abrasion for CRBs with dispersed fillers and hydrofluoric acid etching for PICN materials) followed by silanization could be beneficial for the long-term bond strength of CRBs. The bonding mechanism may include: 1) micromechanical interlocking between CRBs and resins, or 2) chemical bonding established by silane coupling agents between the glassy phase of CRBs and resins [4, 13]. Moreover, the wettability of the conditioned surface is important for bonding of CRBs. The silane coupling agents render the CRBs surface hydrophobic, and the non-hydrolyzable functional groups at the other end of the silane molecules can co-polymerize with the methacrylate groups of resins [15].

Recently, many so-called universal adhesives (UAs) have been developed and introduced into the dental market. UAs generally include a silane coupling agent, an acidic functional monomer (e.g., 10-methacryloyloxydecyl dihydrogen phosphate (10-MDP)), a solvent (e.g., ethanol, water), and/or a methacrylate monomer [16]. 10-MDP has been demonstrated to promote bonding of zirconia, alumina, and metals, which are the components of CRBs [17, 18]. Moreover, chemical bonding to the polymer may be achieved via acid groups of the copolymer and 10-MDP in the UAs

[13]. In this manner, the UAs may establish adhesion to both the polymer and glassy phases of CRBs and provide better performance than that provided by silane coupling agents alone. In a recently published study, UA application without mechanical conditioning showed significantly higher bond strength values of PICN material than those resulting from the other mechanical conditioning methods (alumina air abrasion, Er,Cr:YSGG laser treatment, and hydrofluoric acid etching) [19]. However, limited information is available to compare the effects of UAs and silane coupling agents on CRBs.

Recently, high strength CRBs (HSCRBs, e.g., KZR-CAD HR3 Gammatheta, Yamakin, Kouchi, Japan) were developed and introduced into the market. The HSCRBs were claimed to have a higher flexural strength (approximately 300 MPa according to the manufacturer's internal data) than that of conventional CRBs [20]. Moreover, the fluoride release from fluoridated fillers may inhibit the bacterial adhesion and maintain surface integrity during its long-term service. The abovementioned properties make it a good alternative to its competitor on the market. According to the manufacturer's recommendation, this material needs to be treated with alumina air abrasion and chemical conditioning prior to bonding [20]. Other than the manufacturer's internal data, information concerning this novel material is very limited. Thus, the objective of this investigation was to evaluate the effects of different chemical conditioning methods following alumina air abrasion on the microtensile bond strength (µTBS) of SARs to the HSCRB. The following null hypotheses were tested: 1) that no differences exist among the 3 chemical conditioning methods

in terms of bond strength, surface morphology, surface composition, and surface roughness to the HSCRB; 2) that no differences exist among the 3 SARs in terms of bond strength to the HSCRB; and 3) that the bond strength of the HSCRB remains the same after thermocycling.

2. Materials and methods

One HSCRB (KZR-CAD HR 3 Gammatheta, Yamakin, Kouchi, Japan [KC]) and 3 dual-cure SARs (RelyX Unicem 2, 3M ESPE, St. Paul, MN, USA [RXU]; SA Luting Plus, Kuraray Noritake Dental, Tainai, Japan [SAL]; and G-Cem ONE, GC Corp., Tokyo, Japan [GCO]) were investigated in this study (Table 1).

2.1. Surface conditioning

Eighteen sections (14.5 × 14.5 × 6 mm) were obtained from KC blocks using a low-speed diamond saw (Isomet, Buehler, Lake Bluff, IL, USA) under continuous water cooling. The sections were randomly divided into 3 groups according to the SARs used for bonding. The cementation surfaces of the sections were wet-polished with 600-grit silicone-carbide abrasive paper using a polishing machine (ML-160A, Maruto, Musashino, Japan) under continuous water cooling. They were sandblasted with 50-µm aluminum oxide particles (Hi Aluminas, Shofu, Kyoto, Japan) at 0.2 MPa for 15 s at a distance of 10 mm using a sandblasting device (Jet Blast III, J. Morita, Kyoto, Japan). The specimens were cleaned with phosphoric acid (K-etchant GEL, Kuraray Noritake Dental, Tainai, Japan) for 5 s and then ultrasonically cleaned in distilled water for 1 min. After cleaning, the specimens were air dried using a 3-way syringe for 10 s. Each group was further divided into 3 subgroups according to the chemical conditioning performed on the cementation surfaces (2 sections per subgroup).

Subgroup C (control): no chemical conditioning was performed.

Subgroup UA (universal adhesive): the UA (Bondmer Lightless, Tokuyama Dental, Ibaraki, Japan) was freshly mixed, applied to the cementation surfaces, and then air dried using a 3-way syringe for 5 s.

Subgroup MM (MPTS + MDP): a mixture of γ -methacryloyloxypropyl trimethoxysilane (γ -MPTS) and 10-MDP in equal volumes was freshly prepared, applied to the cementation surfaces, and then air dried using a 3-way syringe for 5 s.

2.2. Cementation procedure

After surface conditioning, 2 sections in the same group (with the same surface conditioning) were cemented together using 1 of the 3 dual-cure SARs tested. Before applying the SARs, two pieces of 2-mm-wide polyethylene adhesive tape (50 μm thick) were attached at the margins of 1 cementation surface as a spacer to standardize the bonding agent/luting cement thickness at 50 μm. The cement was left to set primarily in self-cure mode for the first 2 min at ambient room light, thereby simulating the clinical time needed for cementation prior to actual light curing [21]. Light irradiation was then performed by placing the tip of the LED light-curing unit (power density of 1000 mW/cm²; Pen-cure, J. Morita, Osaka, Japan) on 4 sides for a total of 160 s. The bonded sections were stored in distilled water at 37°C for 24 h prior to specimen preparation for the μTBS test.

2.3. Micro-tensile bond strength test

Forty beams (bonding area: ~0.81 mm²) were cut from each bonded section using a low-speed diamond saw under constant water cooling. The exact bonded area of each

beam was measured with a digital caliper (Mitutoyo, Tokyo, Japan) and further used for the μ TBS calculation.

Per chemical conditioning and per SAR, 40 beams were prepared and tested under 2 different storage conditions (n = 20 per storage condition). Non-aged beams were stored in distilled water at 37°C for 24 h prior to the μ TBS test. The aged beams were subjected to 10,000 thermal cycles between water baths (Rika-Kogyo, Tokyo, Japan) held at 4°C and 60°C, with a dwelling time of 1 min per cycle in each bath [22, 23].

Following storage, the ends of each beam were fixed to a jig with cyanoacrylate glue (Model Repair, Dentsply-Sankin, Tokyo, Japan). The jig was fixed into a universal testing machine (EZ-test short, Shimadzu, Kyoto, Japan) and stressed under tensile force at a crosshead speed of 1 mm/min until failure occurred. The μ TBS (in MPa) was then calculated by dividing the imposed force (in N) at the time of fracture by the bonded area (in mm²). Pre-test failures were recorded when the specimens failed before the actual testing.

The debonded surfaces of all specimens were examined using an optical microscope (SMZ-10, Nikon, Tokyo, Japan) at a magnification of 50x. The mode of failure was categorized as adhesive failure (at the CRB/SAR interface), cohesive failure (within the CRB/SAR), or mixed failure (combination of the two modes on the same surface).

2.4. Contact angle measurement of adherend surfaces

The difference in the hydrophilicity/hydrophobicity of the KC material after surface conditioning was determined by the contact angle measurements, obtained

with a contact angle goniometer (CA-D, Kyowa Interface Science, Niiza, Japan). Distilled water was used as the test liquid. The measurements were performed on the conditioned specimen surfaces using the sessile-drop method at 23 ± 1 °C before and after ethanol cleaning. Briefly, ethanol cleaning was performed by placing the specimens in absolute ethanol solvent in the ultrasonic bath for 2 min to remove any non-chemically-bonded molecules after chemical conditioning [24]. Five sets of contact angle measurements were obtained for each sample, and the right and left contact angles for the droplets were averaged. The size and volume of the drops were kept constant, as variations in the drop volume can lead to inconsistent contact angle measurements [25]. Final contact angle measurements were recorded (usually within 10–15 s) once their values stabilized after drop placement.

2.5. Surface roughness measurement of adherend surfaces

After surface conditioning, the surface roughness of the KC specimens (n = 8 per chemical conditioning) was measured using a surface profilometer (SEF680, Kosaka, Tokyo, Japan). For each measurement, the surface roughness was measured from the profiles obtained by a needle passing across a length of 6 mm at a speed of 0.5 mm/s. The average arithmetical mean roughness (Ra) values (μ m) were calculated.

2.6. Scanning electron microscopy (SEM) observation and energy-dispersive Xray spectroscopy (EDS) analysis of adherend surfaces

After surface conditioning, KC specimens (n=3 per chemical conditioning) were analyzed by scanning electron microscopy (SEM) (Quanta 250, FEI, Hillsboro, OR, USA). The specimens were mounted on aluminum stubs and sputter-coated with gold before being examined at an acceleration voltage of 15 kV. Furthermore, quantitative changes in the adherend surface composition were evaluated using EDS. The respective EDS spectra were obtained over a 400 x 400 μ m area with an acceleration voltage of 15 kV.

2.7. Statistical analysis

The assumption of normality and equality of variances was confirmed with the Shapiro-Wilk and Levene tests, respectively. The µTBS data were analyzed using three-way analysis of variance (ANOVA), the contact angle data were analyzed using two-way ANOVA, and the surface roughness data and EDS data were analyzed using one-way ANOVA. Non-parametric failure mode data were analyzed using Fisher's exact probability test. All statistical analyses were performed using PASW version 18 (SPSS, Chicago, IL, USA) at a significance level of 5%.

3. Results

3.1. SEM and EDS results of adherend surfaces

Representative SEM images are shown in Figure 1. Similar surface morphology was found in the subgroups C (control) and MM (MPTS + MDP). In contrast, the surface of subgroup UA (universal adhesive) was covered by a thin film (Fig. 1). Table 2 presents the concentration (wt%) of the respective elements on the adherend surfaces as detected by EDS. In particular, after surface conditioning with UA (1.23 ± 0.35 wt%) and MPTS + MDP (1.31 ± 0.32 wt%), significant increases in P concentration were found on the surface that corresponds to phosphate.

3.2. Surface roughness measurement of adherend surfaces

The Ra values of all tested adherend surfaces ranged from 2.20 to 3.10 μ m (Table 3). Significant effects were found for the factor "chemical conditioning" (P < 0.001). Subgroup UA (3.10 \pm 0.35 μ m) has demonstrated the highest Ra values, followed by subgroups MM (2.81 \pm 0.17 μ m) and C (2.20 \pm 0.11 μ m).

3.3. Contact angle measurement of adherend surfaces

Significant effects were found for the factors "chemical conditioning" and "ethanol cleaning" (all P < 0.001). Furthermore, there was a significant interaction between chemical conditioning and ethanol cleaning (P < 0.001). The initial water contact angle (before ethanol cleaning) of KC material ranged from 28.42° to 56.16° (Table 4). After ethanol cleaning, the samples from the subgroups UA and MM showed an increase in the contact angle, whereas no changes were found in subgroup C. The

samples from subgroup MM yielded significantly higher water contact angles than those from subgroups UA and C.

3.4. Micro-tensile bond strength test and failure mode analysis

No pre-test failures were observed in the present study. According to the three-way ANOVA of the μ TBS data, significant effects were found for the factors "cement system", "chemical conditioning", and "aging" (Table 5). There were significant two-factor interactions between the cement system and chemical conditioning (P < 0.001) as well as between chemical conditioning and aging (P < 0.001). However, the interaction between the cement system and aging was not significant (P = 0.083). Moreover, there was a significant three-factor interaction among the cement system, chemical conditioning and aging (P < 0.001).

The means and standard deviations of the μ TBS values are listed in Table 6. The μ TBS values in the RXU group were significantly higher than those in the other 2 groups. The μ TBS values of the tested SARs can be ranked as follows: RXU > GCO > SAL. Chemical conditioning following alumina air abrasion prior to bonding with the SARs yielded significantly higher μ TBS values than those of the control under both non-aged and aged conditions. MM application prior to bonding yielded the highest bond strength among the different chemical conditioning methods used. The μ TBS values of the tested chemical conditioning methods can be ranked as follows: MM > UA > C. After thermocycling, the samples in the RXU/MM, RXU/UA, and GCO/MM groups showed no significant changes in the μ TBS values, whereas the others showed a significant reduction.

The failure mode distribution of the different groups is shown in Fig. 2. The predominant failure mode was cohesive failure in all groups. Overall, both chemical conditioning and aging showed a significant effect on the fractography (P = 0.005 and P < 0.001, respectively). The cement system had no influence on the failure mode (P = 0.791). Increased adhesive failures were observed in all groups after aging, while chemical conditioning led to fewer adhesive failures.

4. Discussion

Based on the present findings, the null hypotheses that no differences existed among the 3 chemical conditioning methods and among the 3 SARs in terms of bond strength, surface morphology, surface composition, and surface roughness to HSCRB were rejected. The null hypothesis that the bond strength of HSCRB would remain the same after thermocycling was accepted for the RXU/MM, RXU/UA, and GCO/MM groups.

Silane coupling agents are useful and effective in promoting adhesion between resin and silica-based restorative materials, such as glass ceramic and CRBs [26]. After being activated in an acidic environment, the silane coupling agent undergoes the hydrolysis process and forms strong siloxane (-Si-O-Si-) linkages on the silicabased material surface [27]. Recently, it has been found that the bond strength of the most frequently used silane coupling agent (γ -MPTS) activated by the acidic functional monomer (10-MDP) was approximately two-fold greater than those obtained with acetic acid [28]. When hydrolyzed with 10-MDP, the self-dehydration condensation of y-MPTS could be inhibited, thereby resulting in an increased number of silane molecules bonded to the glassy phases and a higher bond strength. Thus, the increase of Si on EDS was revealed in both subgroups UA and MM, although the increase was not significant (Table 2). Furthermore, 10-MDP is a proven functional adhesive monomer that can chemically bond to zirconia, alumina, and metals [29]. In terms of effects, both subgroups UA and MM revealed an increased wt% in P. The increase of both Si and P on EDS indicates the synergistic effects of y-MPTS and 10-

MDP, which improve the bonding performances of HSCRB. Therefore, both the UA and MM applications yielded a significant increase in the bond strength and fewer adhesive failures than the controls. A high prevalence of cohesive and mixed failures also implied a strong bond to the conditioned HSCRB.

Interestingly, the bond strength of KC material differed between the commercial UA (Bondmer Lightless, Tokuyama Dental) and the mixture of γ -MPTS and 10-MDP. Specifically, in contrast to the samples treated with the UA, the samples treated with the mixture of γ -MPTS and 10-MDP exhibited significantly higher μ TBS values. The UA tested contains γ -MPTS and 10-MDP, as do most commercially available products on the market. Previous studies showed that the γ -MPTS in the UA may become hydrolyzed and deteriorate in a single bottle due to the presence of water with 10-MDP and may not be effective in optimizing bonding to silica-based restorative materials [16, 30]. Moreover, the residual solvent (water) that remains within the adhesive interface may weaken the bond [31]. However, this may not be the case for the UA application since the γ -MPTS and 10-MDP were stored in separate bottles and were mixed immediately before application to the HSCRB surface. The difference in the performances of the UA and the mixture of γ -MPTS and 10-MDP may be related to the following factors: 1) the different effects of the UA and the mixture of γ -MPTS and 10-MDP on the chemical functionalization of the adherend surfaces. Greater effects were associated with the application of γ -MPTS and 10-MDP since the increased surface P concentration was detected by EDS analysis. Only a minor increase in the surface Si concentration was found after surface conditioning, possibly

due to the Si-rich surface of the KC block; 2) as a self-assembled ultra-thin film, the γ -MPTS and 10-MDP coating has a weaker mechanical strength than SAR and HSCRB and can be regarded as the weakest point in the adhesion system [32]. An increase in the coating thickness, as observed on the adherend surfaces after the UA application, may reduce the bonding performance of the chemical conditioning [33]; or 3) organic additives in the UA and the different concentration of γ -MPTS and 10-MDP in the 2 solutions tested may also play an important role [28, 34]. In addition, the samples treated with the UA exhibited different initial contact angles (before ethanol rinsing) than those treated with the mixture of γ -MPTS and 10-MDP. This result clearly demonstrated that the organic additives in the UA are deposited and/or physiosorbed onto the HSCRB surface. Ethanol cleaning removed the non-chemically bonded molecules and therefore led to an increase in the contact angle [24]. The observed lower contact angles with the UA probably indicate a decrease in the amount of chemisorbed γ -MPTS. This could be because the organic additives prevent chemisorption of γ -MPTS [28]. It is also important to note that slightly rougher surfaces were found after the application of UA and the mixture of γ -MPTS and 10-MDP. The changes in surface roughness may affect the surface wettability and contribute to the enhanced bonding performance of the adherend surfaces [35].

The surface of subgroup UA was rougher than that of other subgroups. It should be noted that the UA contains an unknown amount of methacrylates such as Bis-GMA, TEGDMA and HEMA. These methacrylates can adhere to any resin-based dental restorative materials as well as primer (e.g. MTU-6, γ -MPTS or 10-MDP) treated

surfaces. These methacrylates contain C=C and polymerizable to become polymer under certain conditions, such as exposure to an acryl borate catalyst, which is present in liquid B of the UA. Therefore, thicker film layers were revealed by SEM (Fig. 1 C and D), which contribute to the roughness on the substrate surface [36].

The bond strength and failure mode differed between the non-aged and aged samples. The non-aged samples exhibited significantly higher μ TBS values and predominantly suffered cohesive failure. Temperature changes during the thermocycling process may amplify the coefficient of thermal expansion mismatch of the bonded materials, which generates mechanical stresses at the bonded interface, resulting in bonding degradation [37]. Notably, the bond strength of the KC material treated with the mixture of 10-MDP and γ -MPTS and followed by cementation with RXU and GCO remained stable after thermocycling. From a clinical viewpoint, treating the KC materials with alumina air abrasion and a mixture of γ -MPTS and 10-MDP may be advantageous to provide long-term adhesion.

Although there are a wide variety of dental materials available for CAD/CAM restorations, HSCRBs have been proven adequate for veneers, inlays/onlays, and crowns due to their superior aesthetic and mechanical properties [1, 2]. However, bonding to HSCRBs is still challenging. This systematic investigation was carried out to determine the effects of chemical conditioning and SARs on the bond strength of mechanically conditioned HSCRB. It has been recommended that micro-retentive surfaces should be generated by either blasting or hydrofluoric acid etching to improve the bonding properties of CRBs [7, 38]. Therefore, the surfaces of HSCRB

were all treated with alumina air abrasion prior to any chemical conditioning to simulate clinical situations and to adhere to the manufacturer's instructions.

Limited information is available regarding the effects of SARs on the bond strengths of HSCRBs. Of the 3 SARs tested, RXU showed the highest bond strength to HSCRB, which is in accordance with previous studies [39, 40]. This phenomenon may be explained by the fact that RXU has higher filler loading (72 wt%) and larger filler particle sizes than the other SARs tested, which leads to superior mechanical properties of the cement and, thus, stronger engagement of the interlocking on the HSCRB surface treated with alumina oxide particles [40]. Moreover, the superior pHneutralization behavior of RXU may have a positive effect on its mechanical stability during thermocycling [41].

Although this *in vitro* study cannot exactly replicate the intraoral conditions with all individual variations, it provides some insights into the effects of chemical conditioning and SARs on the bonding properties of HSCRB. However, the findings should be interpreted with caution before being applied to clinical situations. Future laboratory and clinical studies are needed to confirm the long-term bonding performance and provide evidence-based guidelines for clinical practice.

5. Conclusion

Within the limitations of this study, the following conclusions can be drawn:

(1) Chemical conditioning with the universal adhesive and a mixture of γ -MPTS and

10-MDP increased the bond strength of HSCRB.

(2) The bond strength varied among the different SARs. RelyX Unicem 2 showed significantly higher μTBS values than SA Luting Plus and G-Cem ONE.

(3) Thermocycling influenced the bond strength of SARs to HSCRBs, while the bond strength was maintained when HSCRB were treated with alumina air abrasion followed by an application of a mixture of γ -MPTS and 10-MDP.

Conflict of interest

The authors declare no conflict of interest.

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Material	Product	Manufacturer	Composition			
Composite resin blocks	KZR-CAD HR 3 Gammatheta	Yamakin, Kouchi, Japan	Methacrylate-based matrix, inorganic filler, pigment			
R Self- G adhesive resins S.	RelyX Unicem 2 Baul, MN, USA		Base paste: methacrylate monomers containing phosphoric acid groups, methacrylate monomers, silanated fillers, initiator components, stabilizers Catalyst paste: methacrylate monomers, alkaline (basic) fillers, silanated fillers, initiator components, stabilizers, pigments			
	G-CEM ONE	GC Corp., Tokyo, Japan	Paste A: fluoro-alumino-silicate glass, UDMA, dimethacrylate, silicon dioxide, initiator, inhibitor Paste B: silicon dioxide, UDMA, dimethacrylate, initiator, inhibitor			
	SA Luting Plus	Kuraray Noritake Dental, Tainai, Japan	 Paste A: 10-MDP, Bis-GMA, triethyleneglycol dimethacrylate, hydrophobic aromatic dimethacrylate, silanated barium glass filler, silanated colloidal silica, Dl-camphorquinone, benzoyl peroxide, initiators Paste B: Bis-GMA, hydrophobic aromatic dimethacrylate, hydrophobic aliphatic dimethacrylate, silanated barium glass filler, silanated colloidal silica, surface treated sodium fluoride, accelerators, pigments 			

Table 1 Materials tested in this study

Universal adhesive	Bondmer Lightless	Tokuyama Dental, Ibaraki, Japan	Liquid A: phosphate monomer, Bis-GMA, TEGDMA, HEMA, MTU-6 Liquid B: acetone, isopropanol, water, acryl borate catalyst, γ-MPTS, peroxide
10-MDP solution	10-MDP solution	GC Corp., Tokyo, Japan	5 wt% 10-MDP in ethanol
Silane solution	γ-MPTS solution	Tokyo Chemical Industry Co., Ltd, Tokyo, Japan	5 wt% γ-MPTS in ethanol

Table 2 Means and standard deviations of the concentration (wt%) of the elements in the different groups determined from EDS

analysis (400 x 400 µm)

Surface conditioning	С	0	F	Na	Al	Si	Sr	Ca	Р	Cl
	24.86	44.31	4.135	0.83	6.50	11.68	7.69			
No conditioning	(1.06) a	(1.00) a	(0.56) a	(0.22) a	(0.83) a	(1.24) a	(0.77) a	-	-	-
TT ' 1 11 '	43.67	39.82	0.41	0.50	0.18	12.68	0.63	0.66	1.23	0.25
Universal adhesive	(0.47) b	(2.97) b	(0.57) b	(0.08) b	(0.04) b	(1.34) a	(0.23) b	(0.06) a	(0.35) a	(0.03) a
WDTC + 10 MDD	30.98	38.96	3.01	0.77	4.19	12.69	8.14		1.31	
γ -MIP I S \pm 10-MDP	(4.65) c	(1.87) b	(0.57) c	(0.01) a	(0.74) c	(0.57) a	(0.88) a	-	(0.32) a	-

Values marked with same lower letter were not significantly different within the column.

"-" = *The concentration of the element was below the detection limit.*

Table 3 Means and standard deviations of the surface roughness (μm) in the

different groups

Surface conditioning	Ra values
No conditioning	2.20(0.11) a
γ -MPTS + 10-MDP	2.81(0.17) b
Universal adhesive	3.10(0.35) c

Values marked with the same lower-case letter were not significantly different within the

column.

Table 4 Means and standard deviations of the water contact angles (°) in the

different groups

Surface conditioning	Before ethanol cleaning	After ethanol cleaning
No conditioning	56.16(4.43) A,a	55.94(4.15) A,a
Universal adhesive	52.84(3.48) A,a	64.73(4.16) B,b
γ -MPTS + 10-MDP	28.42(3.04) A,b	72.55(5.08) B,c

Values marked with the same upper-case letter were not significantly different within each row; values marked with the same lower-case letter were not significantly different within

each column.

	Type III sum		Mean		
Source	of squares	df	square	F	Р
Cement system	10214.358	2	5107.179	42.066	<0.001
Chemical conditioning	41579.034	2	20789.517	171.236	<0.001
Aging	15142.574	1	15142.574	124.724	< 0.001
Cement system X Chemical					
conditioning	3399.432	4	849.858	7.0000	<0.001
Cement system X Aging	610.000	2	305.000	2.512	0.083
Cement system X Chemical					
conditioning X Aging	3921.929	2	1960.964	16.152	<0.001
Chemical conditioning X					
Aging	4472.092	4	1118.023	9.209	<0.001
Error	43707.169	360	121.409		

Table 5 Three-way analysis of μTBS values

		Surface		
Group	Subgroup	conditioning	Non-aged	Aged
	SAL/C	No conditioning	50.75(5.71) A,a	37.41(7.32) B,a
CAL	SAL/UA	Universal adhesive	62.02(9.78) A,b	45.74(13.03) B,b
SAL	SAL/MM	γ-MPTS + 10-MDP	74.49(11.67) A,c	57.64(9.16) B,c
	GCO/C	No conditioning	57.63(10.49) A,a	40.65(10.59) B,a
GCO	GCO/UA	Universal adhesive	63.82(14.42) A,b	56.34(10.25) B,b
	GCO/MM	γ-MPTS + 10-MDP	82.02(11.81) A,c	78.44(9.66) A,c
	RXU/C	No conditioning	68.57(11.43) A,a	35.68(8.13) B,a
RXU	RXU/UA	Universal adhesive	74.89(9.88) A,a	70.29(15.32) A,b
	RXU/MM	γ-MPTS + 10-MDP	76.49(11.47) A,a	75.86(13.24) A,b

Table 6 Means and standard deviations of μTBS values in the different groups

Values marked with the same upper-case letter were not significantly different within each

row; values marked with the same lower-case letter were not significantly different within each column for each cement tested.

Figure 1. Representative SEM images of the adherend surfaces of KC blocks. (A) and (B): subgroup C at 300x and 1,000x; (C) and (D): subgroup UA at 300x and 1,000x; (E) and (F): subgroup MM at 300x and 1,000x.



Figure 2. Failure mode distribution (percentage of each failure mode category) in the different groups

